

# LARGE REDUCTION OF FABRICATION TEMPERATURE FOR FULLY EPITAXIAL Fe/GaO<sub>x</sub>/Fe MAGNETIC TUNNEL JUNCTIONS

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## I. INTRODUCTION

Semiconducting materials have recently attracted considerable attention to the tunnel barrier of MTJs because they provide unique properties and functions to the MTJ such as very low resistance-area product [1] and tunability of a tunneling current by electric fields [2]. Very recently, we have reported a high MR ratio up to 92% in fully epitaxial Fe(001)/GaO<sub>x</sub>(001)/Fe(001) MTJs [3], where the GaO<sub>x</sub> is one of the emerging semiconductors for practical applications. Although GaO<sub>x</sub> is amorphous in the as-grown state, a single-crystalline GaO<sub>x</sub> with a MgAl<sub>2</sub>O<sub>4</sub>-type spinel structure was successfully formed by an *in situ* annealing of the as-grown GaO<sub>x</sub> layer. However, the formation temperature of the single-crystalline GaO<sub>x</sub> is too high (~500°C) to apply to practical applications. In this study, we developed a novel fabrication process that can largely reduce the formation temperature of the fully epitaxial MTJ from 500°C to 250°C.

## II. SAMPLE PREPARATIONS

MTJ films were prepared by molecular beam epitaxy. The structure of the MTJ was Au (10 nm) cap / Co (5 nm) pinned layer / Fe (5 nm) upper electrode / GaO<sub>x</sub> (2 nm) tunnel barrier / MgO (1 nm) seed layer / Fe (30 nm) bottom electrode / MgO (10 nm) buffer layer on MgO(001) substrates. The GaO<sub>x</sub> barrier layer was deposited at 80°C under an O<sub>2</sub> pressure of  $1 \times 10^{-6}$  Torr. Then, an *in situ* annealing at the temperature  $T_{\text{GaO}}$ , where  $T_{\text{GaO}}$  ranges from 250°C to 500°C, was carried out under an O<sub>2</sub> pressure of  $1 \times 10^{-7}$  Torr. The Fe upper electrode was grown and annealed at  $T_{\text{Fe}} = 250^\circ\text{C}$  under the high vacuum below  $1 \times 10^{-9}$  Torr. The  $T_{\text{GaO}}$  and  $T_{\text{Fe}}$  of the present MTJs are listed in Table I.

## III. RESULTS

Figures 1 (a)-(l) show reflection high-energy electron diffraction (RHEED) images of the GaO<sub>x</sub> barrier layers (upper panels), the Fe upper electrode in the as-grown state (middle panels) and after an *in situ* annealing at  $T_{\text{Fe}} = 250^\circ\text{C}$  (bottom panels) of the MTJs, respectively. For the GaO<sub>x</sub> layers, no clear diffraction patterns were observed in the RHEED images for the as-grown state (Fig. 1a) and after the annealing at  $T_{\text{GaO}} = 250^\circ\text{C}$  (Fig. 1b). With increasing  $T_{\text{GaO}}$ , streaky patterns started to appear at around  $T_{\text{GaO}} = 350^\circ\text{C}$  (Fig. 1c), and finally sharp streaky patterns could be observed at  $T_{\text{GaO}} = 500^\circ\text{C}$  (Fig. 1d). These indicate that the GaO<sub>x</sub> barrier layers are amorphous for the samples A and B, mixture of amorphous and crystalline for the sample C and single-crystalline for the sample D, respectively.

The Fe upper electrodes of the samples A and B exhibited broad ring RHEED patterns in the as-grown state (Figs. 1e and 1f), suggesting polycrystalline Fe. In contrast, RHEED images of the samples C and D showed spotty patterns (Figs. 1g and 1h, respectively), implying single-crystalline Fe electrodes. It should be remarked that the broad ring patterns observed in the samples A and B changed to streak ones after an *in situ* annealing at  $T_{\text{Fe}} = 250^\circ\text{C}$  as displayed in Figs. 1(i) and 1(j), respectively. Consequently, the Fe upper electrodes for all the samples revealed similar sharp streak

Table I. Sample name, *in situ* annealing temperatures of GaO<sub>x</sub> barrier ( $T_{\text{GaO}}$ ) and Fe upper electrode ( $T_{\text{Fe}}$ ) for the MTJ samples.

Sample name	$T_{\text{GaO}}$ (°C)	$T_{\text{Fe}}$ (°C)
A	w/o	250
B	250	250
C	350	250
D	500	250

patterns after the *in situ* annealing at  $T_{Fe} = 250^\circ\text{C}$ . This strongly suggests that a single-crystalline Fe upper electrode can be formed even on the as-grown  $\text{GaO}_x$  barrier layer without a high temperature annealing up to  $500^\circ\text{C}$ .

From the RHEED observations, we can expect the existence of coherent spin-polarized tunneling, and thereby a high MR ratio beyond the Julliere’s model even for the samples A and B. Figure 2(a) shows a typical MR curve of sample A. The MR ratio up to 102% was observed at RT, which is close to the reported value in the fully epitaxial MTJ (92%) [3] As plotted in Fig. 2(b), the MR ratio hardly depends on the  $T_{\text{GaO}}$ , suggesting that there is no remarkable difference in the magneto-transport properties among the MTJ samples.

IV. CONCLUSIONS

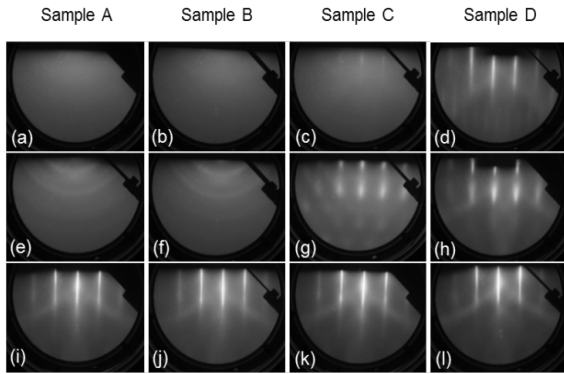
We investigated structural and magneto-transport properties of Fe/ $\text{GaO}_x(\text{MgO})/\text{Fe}$  MTJs grown by different *in situ* annealing conditions for amorphous  $\text{GaO}_x$  tunnel barrier. Fabrication of fully epitaxial MTJ was possible even without the *in situ* annealing of the  $\text{GaO}_x$  barrier, resulting in a large reduction on the formation temperature of the fully epitaxial structure from  $500^\circ\text{C}$  to  $250^\circ\text{C}$ .

ACKNOWLEDGMENTS

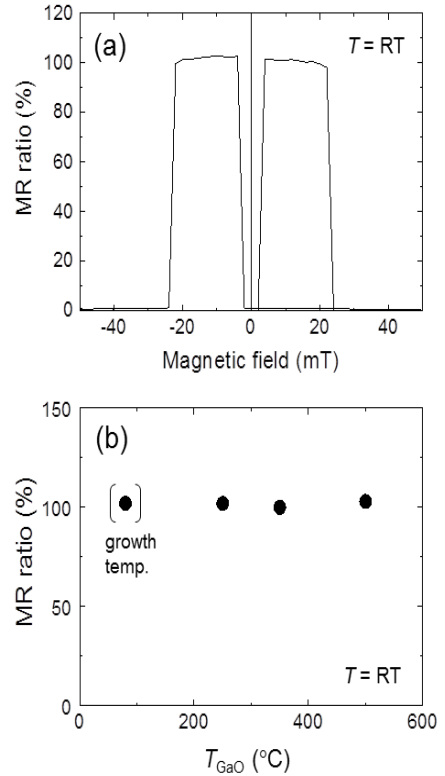
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Figs. 1 RHEED images of the (a)  $\text{GaO}_x$  barrier layer in the as-grown state, (b)-(d) same layer after *in situ* annealing at  $T_{\text{GaO}}$ , (e)-(h) Fe upper electrode in the as-grown state, and (i)-(l) same layer after an *in situ* annealing at  $T_{\text{Fe}}$ , respectively.



Figs. 2 (a) Typical MR curve of sample A and (b) MR ratio as a function of  $T_{\text{GaO}}$  at RT, respectively.