

LARGE REDUCTION OF FABRICATION TEMPERATURE FOR FULLY EPITAXIAL Fe/GaO_x/Fe MAGNETIC TUNNEL JUNCTIONS

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I. INTRODUCTION

Semiconducting materials have recently attracted considerable attention to the tunnel barrier of MTJs because they provide unique properties and functions to the MTJ such as very low resistance-area product [1] and tunability of a tunneling current by electric fields [2]. Very recently, we have reported a high MR ratio up to 92% in fully epitaxial Fe(001)/GaO_x(001)/Fe(001) MTJs [3], where the GaO_x is one of the emerging semiconductors for practical applications. Although GaO_x is amorphous in the as-grown state, a single-crystalline GaO_x with a MgAl₂O₄-type spinel structure was successfully formed by an *in situ* annealing of the as-grown GaO_x layer. However, the formation temperature of the single-crystalline GaO_x is too high (~500°C) to apply to practical applications. In this study, we developed a novel fabrication process that can largely reduce the formation temperature of the fully epitaxial MTJ from 500°C to 250°C.

II. SAMPLE PREPARATIONS

MTJ films were prepared by molecular beam epitaxy. The structure of the MTJ was Au (10 nm) cap / Co (5 nm) pinned layer / Fe (5 nm) upper electrode / GaO_x (2 nm) tunnel barrier / MgO (1 nm) seed layer / Fe (30 nm) bottom electrode / MgO (10 nm) buffer layer on MgO(001) substrates. The GaO_x barrier layer was deposited at 80°C under an O₂ pressure of 1×10^{-6} Torr. Then, an *in situ* annealing at the temperature T_{GaO} , where T_{GaO} ranges from 250°C to 500°C, was carried out under an O₂ pressure of 1×10^{-7} Torr. The Fe upper electrode was grown and annealed at $T_{\text{Fe}} = 250^\circ\text{C}$ under the high vacuum below 1×10^{-9} Torr. The T_{GaO} and T_{Fe} of the present MTJs are listed in Table I.

III. RESULTS

Figures 1 (a)-(l) show reflection high-energy electron diffraction (RHEED) images of the GaO_x barrier layers (upper panels), the Fe upper electrode in the as-grown state (middle panels) and after an *in situ* annealing at $T_{\text{Fe}} = 250^\circ\text{C}$ (bottom panels) of the MTJs, respectively. For the GaO_x layers, no clear diffraction patterns were observed in the RHEED images for the as-grown state (Fig. 1a) and after the annealing at $T_{\text{GaO}} = 250^\circ\text{C}$ (Fig. 1b). With increasing T_{GaO} , streaky patterns started to appear at around $T_{\text{GaO}} = 350^\circ\text{C}$ (Fig. 1c), and finally sharp streaky patterns could be observed at $T_{\text{GaO}} = 500^\circ\text{C}$ (Fig. 1d). These indicate that the GaO_x barrier layers are amorphous for the samples A and B, mixture of amorphous and crystalline for the sample C and single-crystalline for the sample D, respectively.

The Fe upper electrodes of the samples A and B exhibited broad ring RHEED patterns in the as-grown state (Figs. 1e and 1f), suggesting polycrystalline Fe. In contrast, RHEED images of the samples C and D showed spotty patterns (Figs. 1g and 1h, respectively), implying single-crystalline Fe electrodes. It should be remarked that the broad ring patterns observed in the samples A and B changed to streak ones after an *in situ* annealing at $T_{\text{Fe}} = 250^\circ\text{C}$ as displayed in Figs. 1(i) and 1(j), respectively. Consequently, the Fe upper electrodes for all the samples revealed similar sharp streak

Table I. Sample name, *in situ* annealing temperatures of GaO_x barrier (T_{GaO}) and Fe upper electrode (T_{Fe}) for the MTJ samples.

Sample name	T_{GaO} (°C)	T_{Fe} (°C)
A	w/o	250
B	250	250
C	350	250
D	500	250

patterns after the *in situ* annealing at $T_{\text{Fe}} = 250^\circ\text{C}$. This strongly suggests that a single-crystalline Fe upper electrode can be formed even on the as-grown GaO_x barrier layer without a high temperature annealing up to 500°C .

From the RHEED observations, we can expect the existence of coherent spin-polarized tunneling, and thereby a high MR ratio beyond the Julliere's model even for the samples A and B. Figure 2(a) shows a typical MR curve of sample A. The MR ratio up to 102% was observed at RT, which is close to the reported value in the fully epitaxial MTJ (92%) [3] As plotted in Fig. 2(b), the MR ratio hardly depends on the T_{GaO} , suggesting that there is no remarkable difference in the magneto-transport properties among the MTJ samples.

IV. CONCLUSIONS

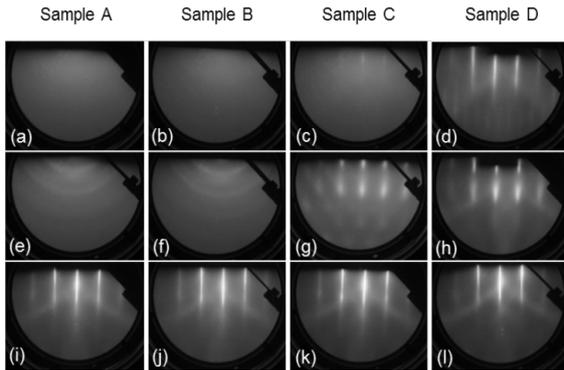
We investigated structural and magneto-transport properties of $\text{Fe}/\text{GaO}_x(\text{MgO})/\text{Fe}$ MTJs grown by different *in situ* annealing conditions for amorphous GaO_x tunnel barrier. Fabrication of fully epitaxial MTJ was possible even without the *in situ* annealing of the GaO_x barrier, resulting in a large reduction on the formation temperature of the fully epitaxial structure from 500°C to 250°C .

ACKNOWLEDGMENTS

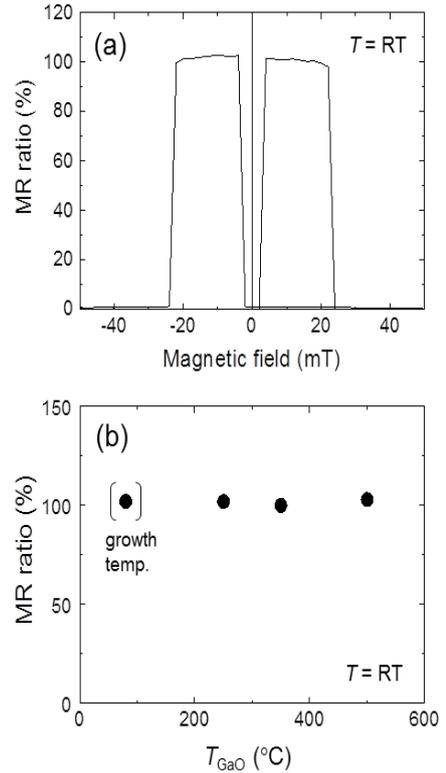
This work was supported by the ImpACT Program of the Council for Science, Technology and Innovation (Cabinet Office, Government of Japan) and Grant-in-Aid for Scientific Research on Innovative Area, "Nano Spin Conversion Science" (Grant No. 26103003).

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Figs. 1 RHEED images of the (a) GaO_x barrier layer in the as-grown state, (b)-(d) same layer after *in situ* annealing at T_{GaO} , (e)-(h) Fe upper electrode in the as-grown state, and (i)-(l) same layer after an *in situ* annealing at T_{Fe} , respectively.



Figs. 2 (a) Typical MR curve of sample A and (b) MR ratio as a function of T_{GaO} at RT, respectively.